



Preparation and Study of the Effect of Annealing Temperature on the Structural, Optical, and Morphological Properties of Nanocrystalline SnO₂ Doped with Cu

Nada K. Abass*, Duha S. Shaker

Physics Department, College of Science for Women, University of Baghdad – Iraq

Article information

Article history:

Received: August, 11, 2021

Accepted: September, 07, 2021

Available online: October, 20, 2021

Keywords:

Cu doped tin oxide,
Spray pyrolysis,
Absorption coefficient,
Transmittance

*Corresponding Author:

Nada K. Abass

nadaka_phys@csw.uobaghdad.edu.iq

DOI:

<https://doi.org/10.53523/ijoirVol8I2ID64>

Abstract

In this study, pure SnO₂ nanocrystalline films were doped with copper using the spray pyrolysis technique. SnCl₂.2H₂O, CuCl₂.2H₂O were used as precursors. The preparation was done in the form of nanoparticles by chemical precipitation method. The prepared materials were annealed at 300°C and 500°C for 1 h to improve crystallization. XRD results of the samples prepared by spray pyrolysis of a solution containing nanoparticles showed that the samples were crystallized in the rutile tetragonal phase. The average crystal size of SnO₂ annealed at 300°C is 3.36 and 3.37 nm for pure and doped samples, respectively, and it is 4.1 nm and 9.75 nm for pure and doped annealed at 500°C, respectively. It is noticed that the crystal structure of SnO₂ does not change with the addition of copper, and the studies of Field Emission Scanning Electron Microscopy confirmed the results where the grain size was within the range (20-50) nm, and the thickness of the films obtained from this assay was in the range (0.9-1.15) μm, with the thickness of doped films at 500°C are higher than those at 300°C. The Atomic Force Microscopy results showed that the roughness rate of the pure films annealed at 300°C and 500°C is 7.99 and 17.4 nm, respectively, while roughness for doped annealed samples were 9.09 and 7.12 nm, respectively. The optical results obtained from UV-Vis analysis showed that the optical bandgap at 300°C for pure and doped samples was (3.40 and 2.8) eV, respectively, while it was (3.75 and 2.59) eV at 500°C for pure and doped samples, respectively. The transmittance decreases with increasing annealing temperature, because the absorbance increases. The extinction coefficient increases, while refractive index decreases with increasing annealing temperature. The absorbance was 0.94 and 1.17 for pure and doped samples at 300°C, and was 1.16 and 1.46 at 500°C.

1. Introduction

Researchers focused on the study of semiconductors in the early nineteenth century, because these materials have impressive properties, such as conductivity change with heat, light, and magnetic fields. The properties of

semiconductor materials have played a significant role in technological applications [1]. Solid crystalline materials are divided according to their ability to conduct electricity into conductors, insulators, and semiconductors (and this classification is based on the energy gap). At absolute zero (0K), semiconducting materials are completely insulators with no electrical conductivity, and as the temperature rises, the electrical conductivity of the materials increases, and it is one of the important characteristics that are used in various technological applications [2]. According to literature, SnO₂ nanofilms belong to a class of materials that combine high optical transparency with good electrical conductivity [3]. Tin oxide is a crystalline solid with a rutile tetragonal structure. It has a large bandgap. n-type in nature with non-stoichiometric structure. The conductivity of SnO₂ thin films can be changed to the best with appropriate dopants [4]. Doped SnO₂ thin films have a bulk energy bandgap of 3.6 eV [5, 6, 7]. These properties make SnO₂ thin films suitable for a variety of applications in many fields of research and device fabrication. The thin-film applications in environmental monitoring through gas sensing is the most essential. SnO₂ properties for gas sensing applications have been improved through catalytic and impurity doping to improve performance [8, 9].

Many methods have been used to deposit SnO₂ based thin films, including RF magnetron sputtering [10], chemical vapor deposition [11], electron beam evaporation [12], flash evaporation [13], dip coating [14], and spray pyrolysis [15]. Spray pyrolysis was chosen for this study because it allows a good control on the grain size and on the atomic-scale growth by varying the precursor content, the temperature of the substrate, and the pressure of the carrier gas [16]. Furthermore, spray pyrolysis technique is inexpensive and effective for large area coating, and it has a good reproducibility; as a result, the method can be used for the production of high-quality thin films on a large scale [17]. In this research, we will prepare nanostructures of SnO₂ and study its physical properties, and then study the effect of doping with copper in different proportions, and choose the best doping ratio that gives the best properties of the prepared material, and study the effect of annealing on it for the possibility of using it in various electronic applications such as solar cells and gas sensors.

2. Experimental Procedure

SnCl₂.2H₂O, CuCl₂.2H₂O, Triton X-100 as a surfactant, ammonia as a precipitant were used to prepare pure and Cu doped tin oxide nanoparticles, which are used to prepare nano-thin films.

Preparation of Pure and Cu Doped Tin Oxide Thin Film

Spray Pyrolysis technique was used to deposit nano-thin films from SnO₂ because it is a simple technique and contains low-cost equipment. The thickness of the films can be controlled by changing the spray coefficients. A solution containing 0.5 M of SnCl₂.2H₂O dissolved in 50 ml of deionized water and mixed with Triton X-100 surfactant at a rate of 0.05ml and ammonia of pH = 8 were mixed. Before spraying the solution, the substrate slides were thoroughly cleaned by ultrasonic bath, followed by ethanol and acetone, then dried well before cutting into 2.5cm × 2.5cm dimensions. After substrate preparation, it was placed onto the electric heater of the spray pyrolysis system, then left for 1h to reach the required temperature for the substrate (400°C). The prepared solution is sprayed onto the substrate using a spraying device, where the distance between the nozzle and the substrate must be taken into account (30±1) cm, the spray time is 5 s, with a stopping time of 55s. The spray rate is 3 ml/min, and the air pressure is 3.5 bar, the substrate temperature is 400°C. After spraying the solution on the slides, the process was repeated several times. Afterward, the electric heater was turned off and the slides were left to cool down to room temperature. In the same way, the solution was prepared, but by adding CuCl₂.2H₂O at a ratio of 2%wt, and then the solution was sprayed on the slides several times. After the slides cooled, they were annealed at 300°C and at 500°C for 1h. Thus, pure SnO₂ and SnO₂:Cu nano-films were obtained.

Characterization

The X-ray diffraction (XRD) pattern of tin oxide nanoparticles was recorded and compared with the standard Card No.96-900-9083 using CuK α , $\lambda = 0.154$ nm. The particle morphology and size were examined by scanning electron microscopy (SEM) ZEISS model SigmaVP with a magnification of 50.00KX and (EDX) Oxford instruments in the UK. Optical properties were analyzed using ultraviolet diffusion. Reflection Spectroscopy was performed using UV-VIS (Shimadzu) within the wavelength range of 280 nm-900nm. AFM was utilized to examine the surface morphology of SnO₂ nanocrystalline thin films. The average grain size was calculated using the cumulative distribution plot.

3. Results and Discussion

XRD Analysis

The significance of this measurement is to know the crystal structure of the materials, indicating the phases of the precipitated materials, and hence the arrangement of the atoms in them and their orientation. The results of the X-ray diffraction examination of the prepared SnO₂ nanoscale films is shown in Figure (1). The figure shows that SnO₂ thin film has a tetragonal type polycrystalline structure for both pure and 2% doped with Cu, where the atomic growth was shown by the dominant crystalline trends (110), (101), (211) and that the prevailing trend is (110) consistent with the Karthick *et. al.* [18]. We can observe some changes in the intensity of the diffraction peaks for some levels after deflection. The peak intensity of doped films is greater than that of pure films, which means that Cu and SnO₂ atoms scatter the X-rays falling on them in the same phase, where the intensity of the peaks decreased, The FWHM value decreases, as well with doping. The crystal size increased according to Scherrer equation [19].

$$D = K\lambda / \beta \cos\theta \quad (1)$$

where, β : is Full Width at half Maximum, θ : is Bragg's angle, and $K = 0.9$. At 300°C annealing temperature, the films exhibit less crystallinity than those annealed at 500°C, because the thermal treatment increases the crystallization regularity, which means that the crystallization of the film material increases and the crystalline defects decrease at 500°C. As well as, the particles become less agglomerated and the atoms are evenly distributed on the sample's surface. At 500°C annealing, the crystal size becomes larger. Tables (1 & 2) show a direct relationship between the temperature of annealing and the size of the nanoparticles [20, 21]. The rise in crystallinity leads to an improvement in the crystal structure. Thus, there are additional peaks that are not present at 300°C. At 500°C, 2θ peaks = 26.56°, 34.05°, 38°, 51.83°, 54.75°, and 62° appeared for the hkl (110), (101), (200), (211), (220), and (310), respectively as shown in Figure (2). Therefore, crystallization of the material increases with the increase in roughness.

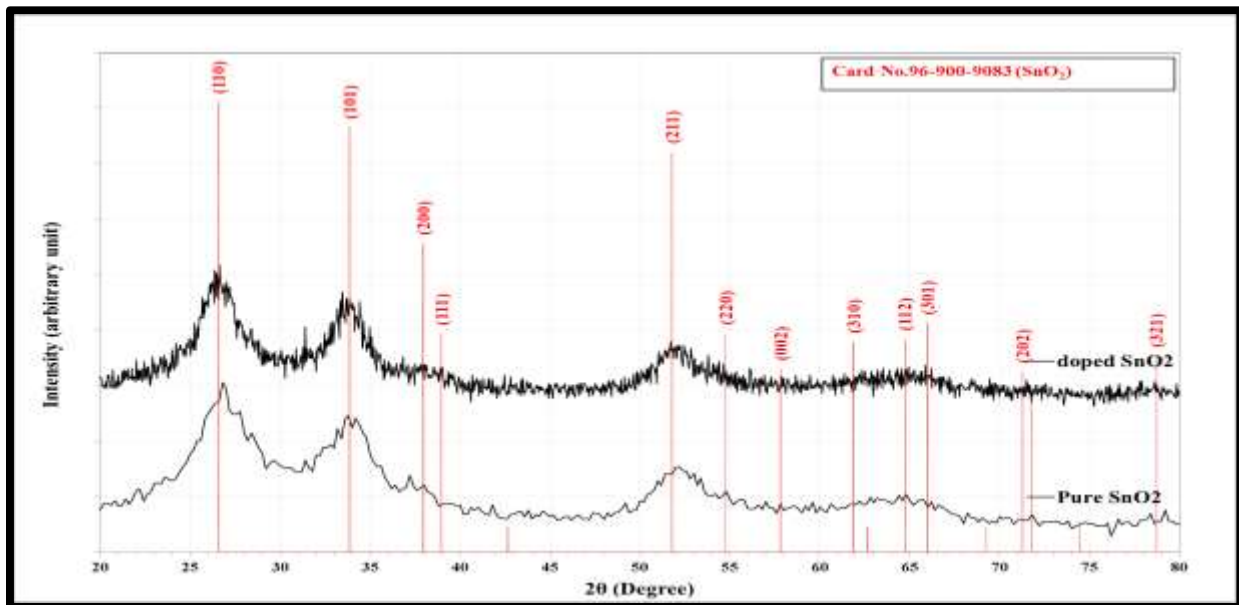


Figure (1). X-ray diffraction pattern of pure SnO₂ & 2%wt doped SnO₂:Cu nanofilm by SPM for annealing temperature of 300° C.

Table (1). XRD Pattern of SnO₂ and 2% wt SnO₂:Cu nano films by SPM for annealing temperature of 300°C.

Sample	2θ (Deg.)	FWHM (Deg.)	d _{hkl} Exp.(Å)	G.S (nm)	hkl	d _{hkl} Std.(Å)	Phase	Card No.
Pure	26.8671	2.6770	3.3157	3.1	(110)	3.3498	SnO ₂	96-900-9083
	33.8506	2.2114	2.6459	3.8	(101)	2.6439	SnO ₂	96-900-9083
	52.1242	2.7353	1.7533	3.2	(211)	1.7642	SnO ₂	96-900-9083
Doped	26.5761	2.3279	3.3514	3.5	(110)	3.3498	SnO ₂	96-900-9083
	33.7342	1.9787	2.6548	4.2	(101)	2.6439	SnO ₂	96-900-9083
	51.8914	2.5606	1.7606	3.5	(211)	1.7642	SnO ₂	96-900-9083

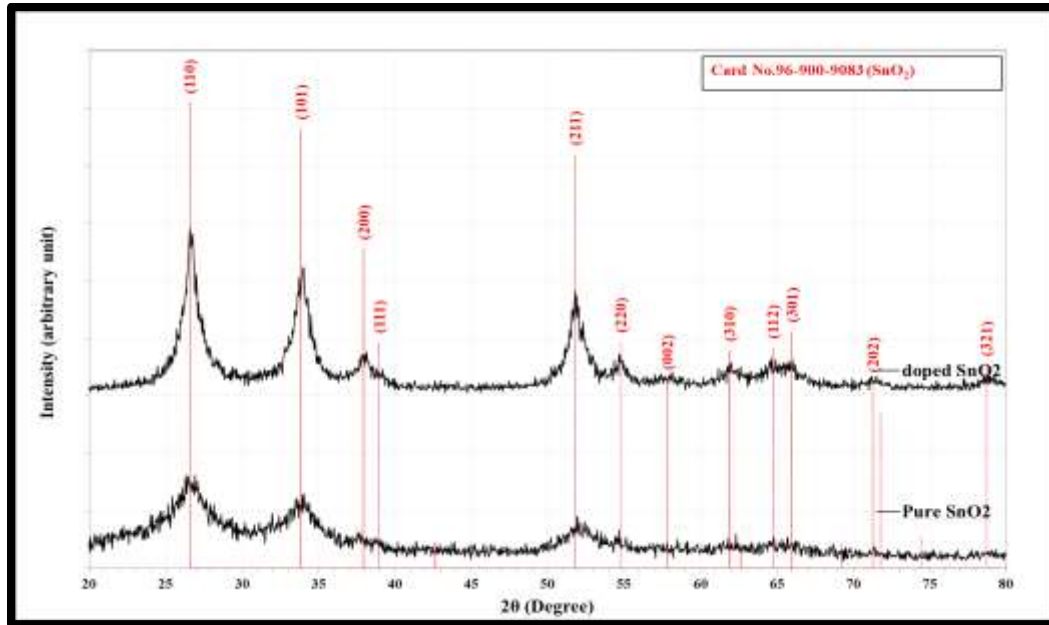


Figure (2). X-ray diffraction pattern of pure SnO₂ & 2%wt SnO₂:Cu nanofilms by SPM for annealing temperature of 500°C.

Table (2). XRD Pattern of SnO₂ and 2% wt SnO₂:Cu nanofilms by SPM for annealing temperature of 500°C.

Sample	2θ (Deg.)	FWHM (Deg.)	d _{hkl} Exp.(Å)	G.S (nm)	hkl	d _{hkl} Std.(Å)	Phase	Card No.
Pure	26.5907	2.1450	3.3496	3.8	(110)	3.3498	SnO ₂	96-900-9083
	33.9275	2.0518	2.6401	4.0	(101)	2.6439	SnO ₂	96-900-9083
	52.0207	1.9586	1.7565	4.5	(211)	1.7642	SnO ₂	96-900-9083
Doped	26.6528	0.9948	3.3419	8.2	(110)	3.3498	SnO ₂	96-900-9083
	34.0518	1.0259	2.6308	8.1	(101)	2.6439	SnO ₂	96-900-9083
	38.0000	0.9327	2.3660	9.0	(200)	2.3686	SnO ₂	96-900-9083
	51.8342	0.9637	1.7624	9.2	(211)	1.7642	SnO ₂	96-900-9083
	54.7565	0.7150	1.6751	12.5	(220)	1.6749	SnO ₂	96-900-9083
	62.0000	0.8083	1.4956	11.5	(130)	1.4981	SnO ₂	96-900-9083

Energy Dispersive X-ray Spectroscopy (EDX)

The EDX results show that the fabricated samples are SnO₂ and SnO₂:Cu, as shown in Figure (3). The quantity of element growth on the sample is shown in Table (3), which includes copper (Cu), tin (Sn), and oxygen (O). Copper functions as a dopant in tin oxide, affecting the sample's transparency and sheet resistance.

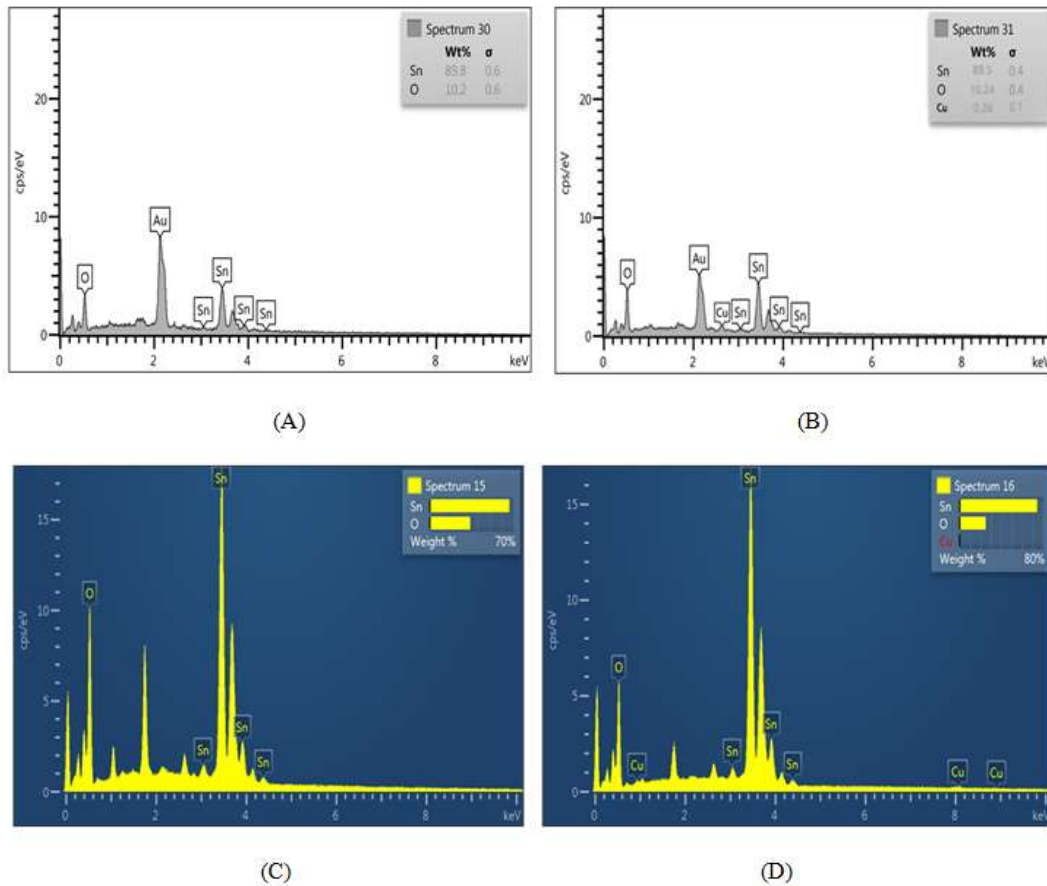


Figure (3). EDX image of (A): pure SnO₂ nanocrystalline at 300°C, (B): SnO₂:Cu at 300°C, (C) pure SnO₂ at 500°C, and (D) SnO₂:Cu at 500°C.

Table (3). The actual composition of the rated EDX prepared materials are scheduled by spray pyrolysis method.

EDX Composition	Temperature	Sn (wt%)	O (wt%)	Cu (wt%)
Pure	300°C	89.4	8.9	0
2% Cu	300°C	82.1	5.6	0.5
Pure	500°C	66.18	33.82	0
2% Cu	500°C	74.46	25.16	0.38

Atomic Force Microscopy (AFM) for Pure SnO₂ Cu Doped SnO₂ Prepared by Spray Pyrolysis Method

The 2D and 3D images show that the SnO₂ topography for all samples has spherical shapes as shown in Figure (4). Moreover, in the area that has been investigated by AFM, it is found that the roughness average of pure SnO₂ NPS and SnO₂:Cu at 300°C annealing are 7.99 nm and 9.09 nm, while the grain size is 56.36 nm and 71.49 nm, respectively. Where the grain size increased in doped samples because crystallinity defects occurred, which leads to an increase in the size due to the addition of Cu, as well as increasing roughness, as shown in the Table

(4). As for at 500°C, we notice that the average roughness decreases when doping with an increase in the size of the grain, as shown in Table (5), because the diameters of the particles are close to each other than the diameters of the different particles (the diameters of the particles are similar) so the roughness is less in doping, but in the pure, the diameters of the particles are different, so the roughness is higher , and this is a guide to the shape of granularity cumulation distribution report in Figure (5).

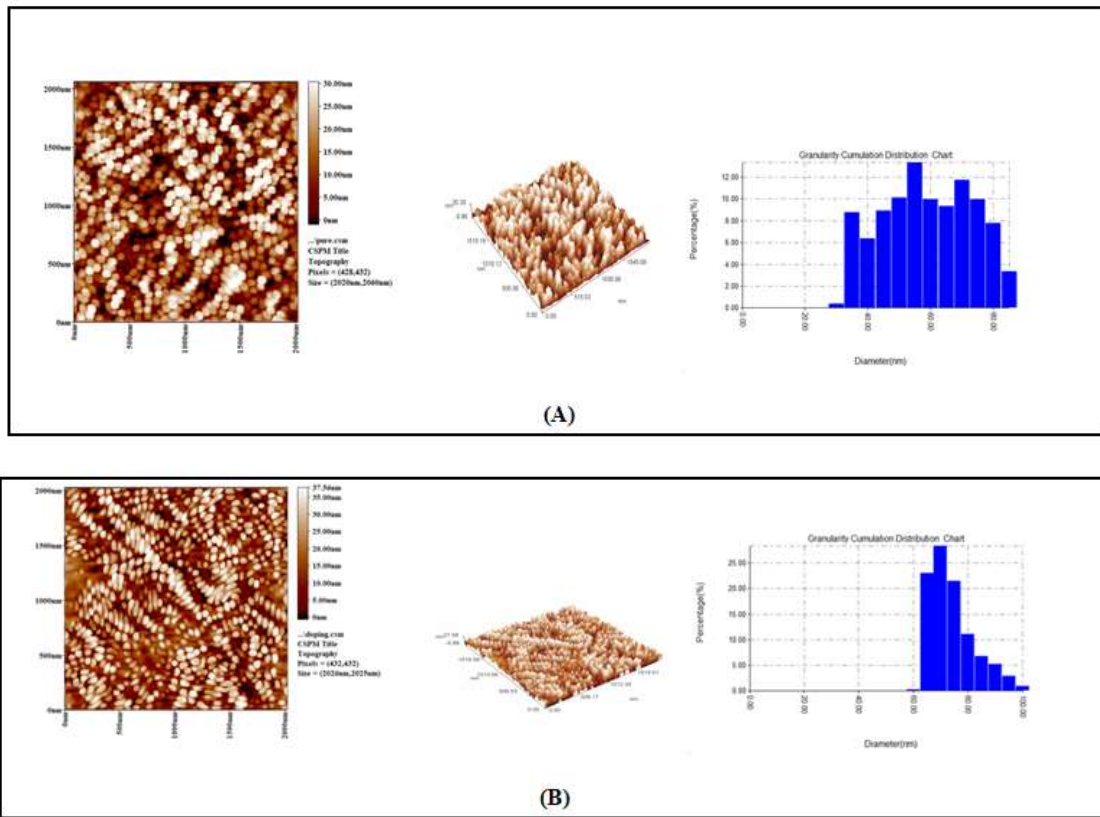


Figure (4). The AFM of (A): pure SnO₂ Nano crystalline in 2D and 3D and Granularity Cumulation Distribution Report, and (B) SnO₂:Cu in 2D and 3D and Granularity Cumulation Distribution Report, annealing at 300°C.

Table (4). The grain size and roughness of pure SnO₂ nanocrystalline and doped with Cu, annealing at 300°C.

Cu Doping Concentration W%	Grain Size (nm)	Roughness (nm)
0	56.36	7.99
2	71.49	9.09

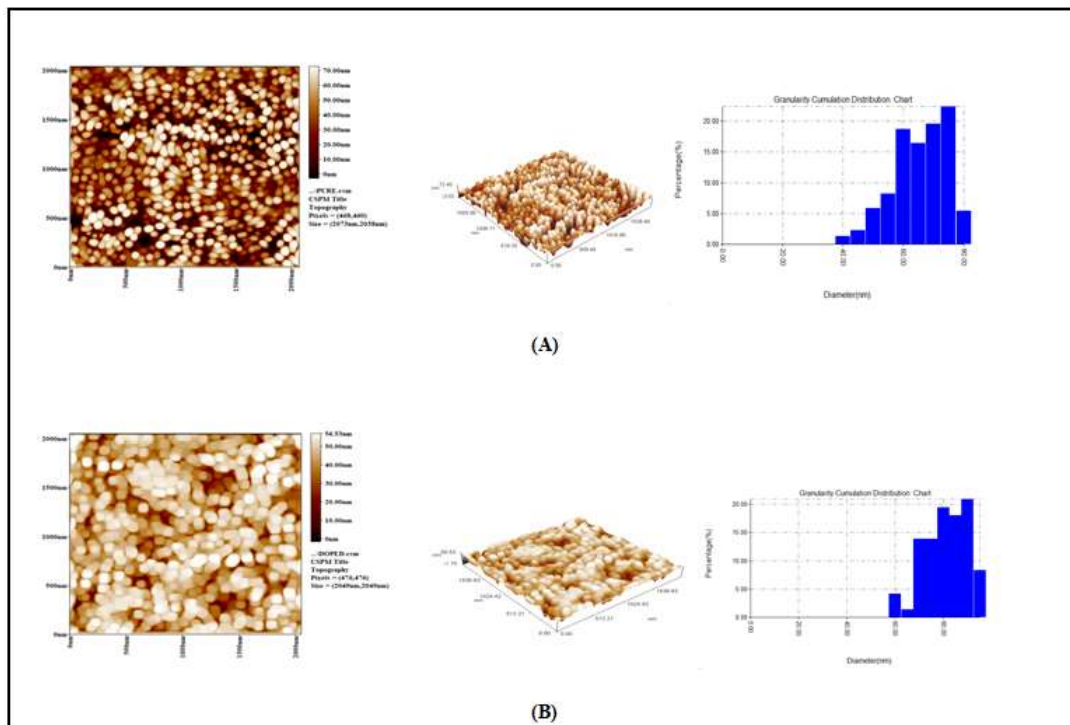


Figure (5). The AFM of (A) pure SnO₂ Nano crystalline in 2D and 3D and Granularity Cumulation Distribution Report, and (B) SnO₂:Cu in 2D and 3D and Granularity Cumulation Distribution Report, annealing at 500°C.

Table (5). The grain size and roughness of pure SnO₂ nanocrystalline and doped with Cu, annealing at 500°C.

Cu Doping Concentration W%	Grain Size (nm)	Roughness (nm)
0	63.15	17.4
2	78.41	7.12

Field Emission Scanning Electron Microscopy (FE-SEM) for Pure Nano crystalline of SnO₂ and Cu Doped SnO₂ Prepared by Spray Pyrolysis Method

By examining FE-SEM images of pure SnO₂ and SnO₂:Cu samples at 2% wt concentration and annealing at 300°C and 500°C as shown in Figure (6), the form and size of the grains can be determined. We observed that samples annealed at 300°C exhibit a size from 40 nm to 73 nm with a spherical shape, proving that SnO₂ nanocrystalline has a nanoparticle domain [21]. The grain size decreasing as the annealing temperature increased. From 18.5nm to 34.5nm at 500°C. The size of the grains reduces as the annealing temperature rises because the annealing process and the resulting deformation effect the recrystallization process, and that grain size increases at 300°C but reduces at 500°C. Grain as well as hardness increase [23, 24]. The thickness of the annealed nano-thin films is measured at 300°C and 500°C, as shown in Figure (7), and it is found that the thickness of the thin films increases as the temperature rises. To enhance grain size, the thickness of the film was enhanced by raising the particle size and increasing the annealing temperature [25].

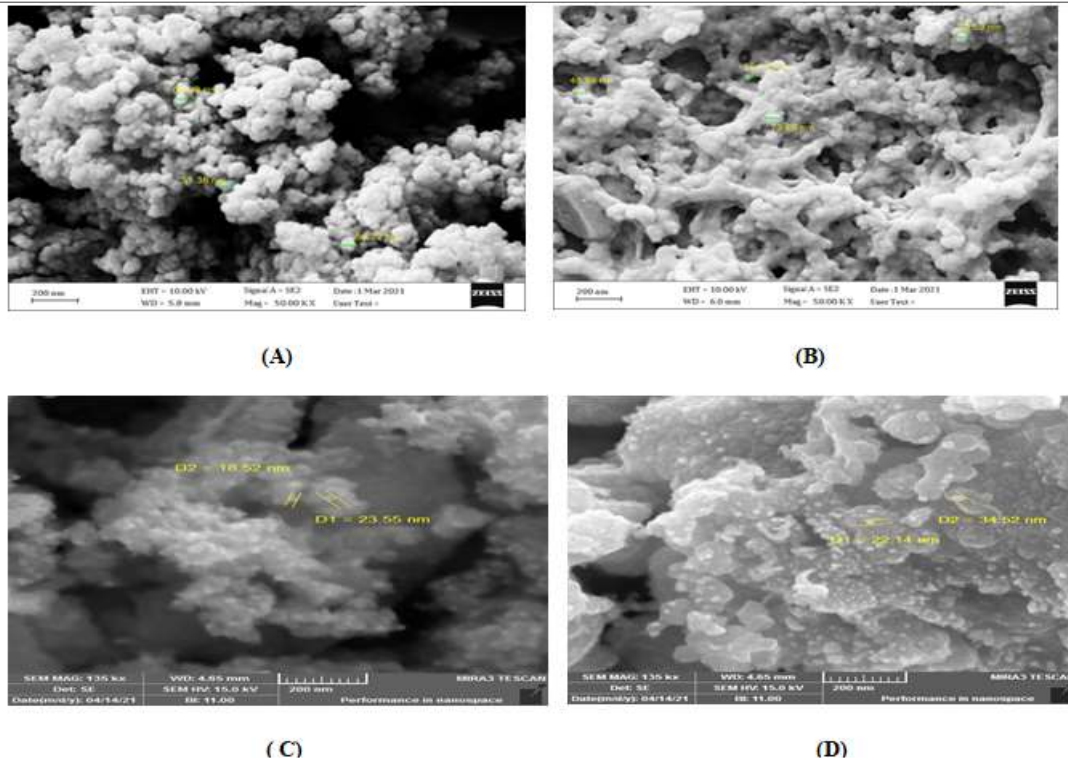


Figure (6). FE-SEM images of (A): pure SnO₂ at 300°C annealing, (b): 2% SnO₂:Cu at 300°C annealing, (C): Pure SnO₂ at 500°C annealing, and (D): 2% SnO₂:Cu at 500°C annealing.

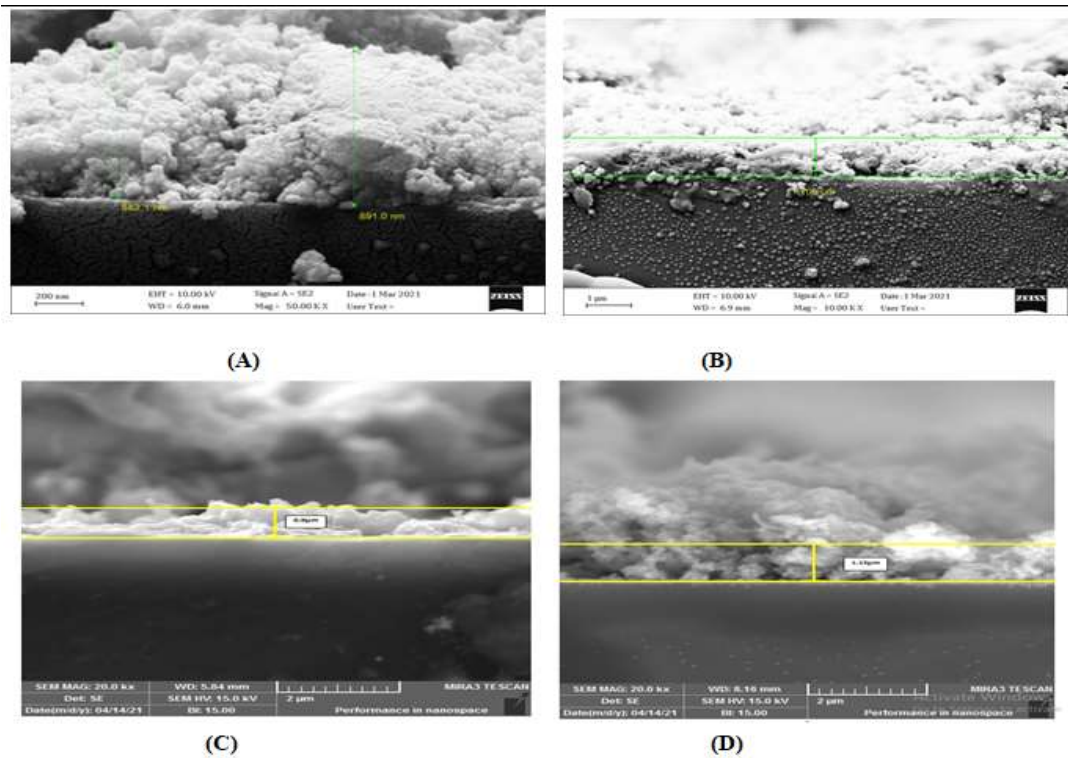


Figure (7). FE-SEM images of Cross Section (A): pure SnO₂ at 300°C annealing, (b): 2% SnO₂:Cu at 300°C annealing, (C): pure SnO₂ at 500°C annealing, and (D): 2% SnO₂:Cu at 500°C annealing.

Optical Measurements of Nanocrystalline Thin Films Prepared by Spray Pyrolysis Method

Figure (8) shows the change in the absorbance spectrum as a function of wavelength and annealing temperature. The absorbance decreases as the annealing temperature increases, and with increasing the wavelength. This means that the incident photon is unable to excite the electron and transfer it from the valence to the conduction band, because its energy is less than the energy gap's value [26]. The number of layers was increased throughout the spraying process to achieve a uniform and homogeneous layer, resulting in a decrease in layer transparency due to an increase in grain size. The layer gets more opaque as the number of layers' increases (Ray *et. al.*) These results agree with AFM results shown elsewhere, which showed that the increasing doping by 2% results in grain size increase at 300°C and 500° C [27], that the transmittance of thin films when annealed at 300°C is higher than that at 500°C as shown in Figure (9), because the thickness of the annealed films at 500°C is greater than the thickness of the annealed films at 300°C and because the annealing worked to reduce crystal defects. As shown in Figure (10), the Extinction Coefficient (K) is calculated for non-annealed and annealed thin films at 300°C and 500°C. Extinction Coefficient varies with photon energy as the energy of the photon increases, and it is influenced by the thickness of the film, that the values of the coefficient of extinction generally increases with the increase of the doping ratio due to the density. The levels formed by doping within the energy gap.

As shown in Figure (11), the refractive index (n) varies with the energy of a photon across a range of wavelengths. The wavelength ranges from 300nm to 1100nm, with n decreasing with increasing film thickness and annealing temperature. This figure shows that the refractive index decreases with higher doping ratio.

Figure (12) shows the values of the energy gap for the permissible direct transmission of the pure and 2%wt doped SnO₂ thin films. . The values of the energy gap decrease with doping and the reason for that is due to the fact that the doping leads to the addition of new sublevels near the valence band. The optical energy gap is critical in determining the feasibility of using Nano-thin films in cell manufacturing. The optical absorption can be utilized in solar cells (photovoltaic), display, and other applications, in which the film is transparent to radiation with an energy less than the energy gap ($h\nu < E_g$). And absorbing radiation with an energy greater than ($h\nu > E_g$), and there are many factors that influence the energy gap.

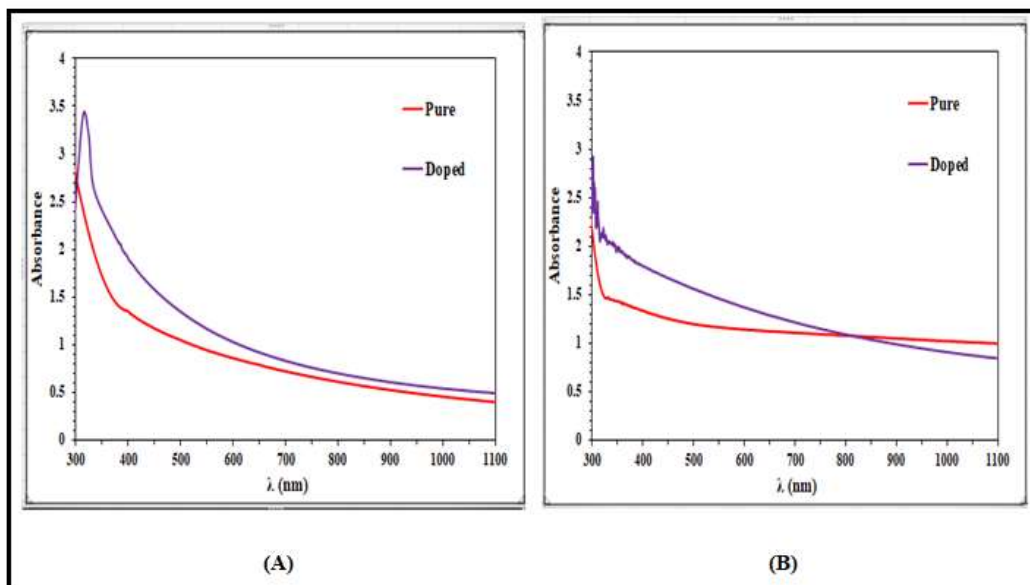


Figure (8). Shows the absorbance spectrum of (A): pure SnO₂ nano crystalline & SnO₂:Cu annealed at 300°C, and (B): pure SnO₂ Nano crystalline & SnO₂:Cu annealed at 500°C.

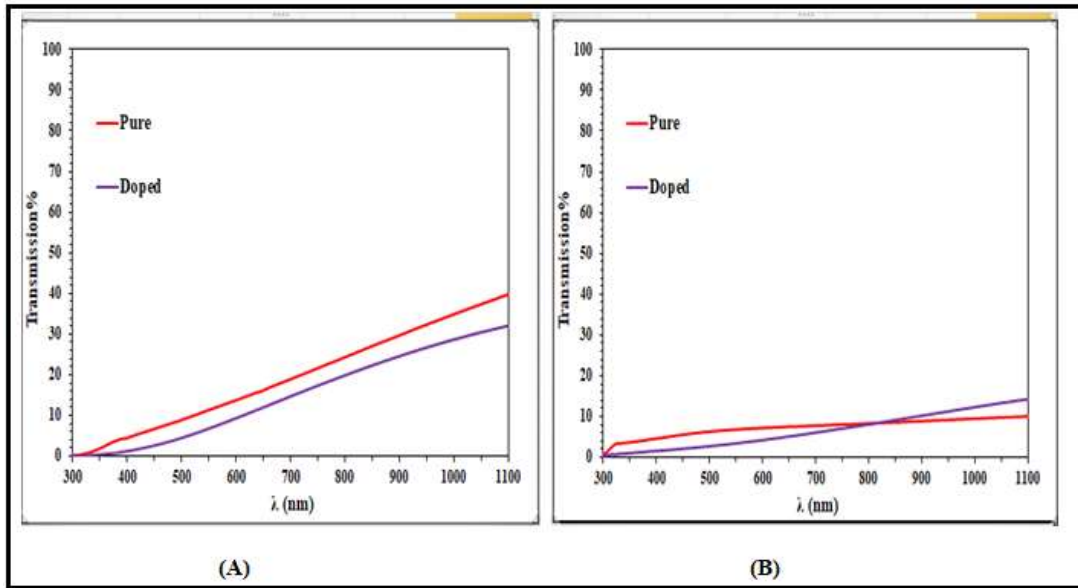


Figure (9). Shows the Transmittance spectrum of (A): pure SnO₂ nanocrystalline & SnO₂:Cu annealed at 300°C, and (B): pure SnO₂ NPs & SnO₂:Cu annealed at 500°C.

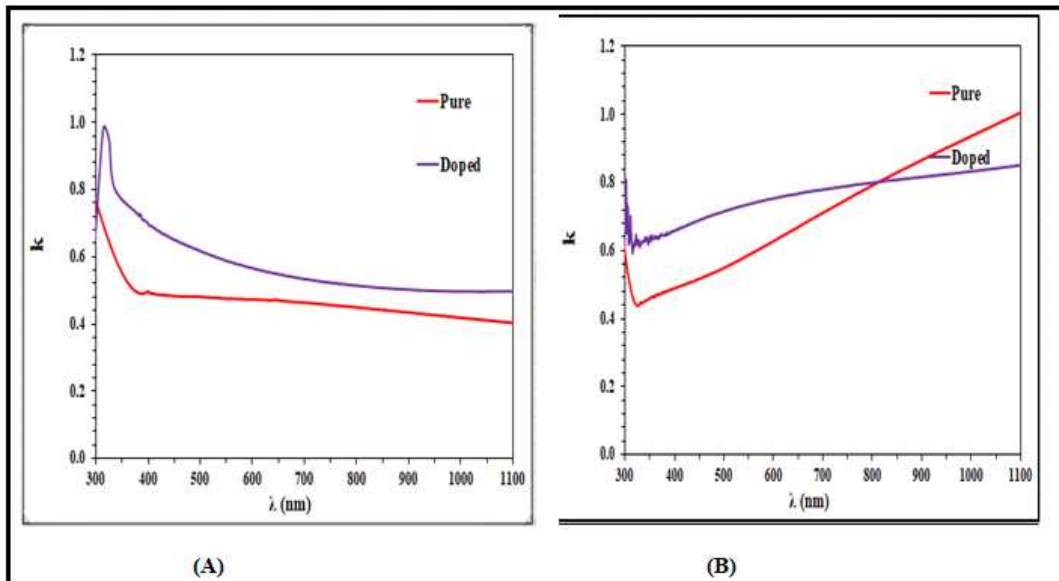


Figure (10). Shows the Extinction Coefficient of (A): pure SnO₂ nanocrystalline & SnO₂:Cu annealed at 300°C, and (B): pure SnO₂ nanocrystalline & SnO₂:Cu annealed at 500°C.

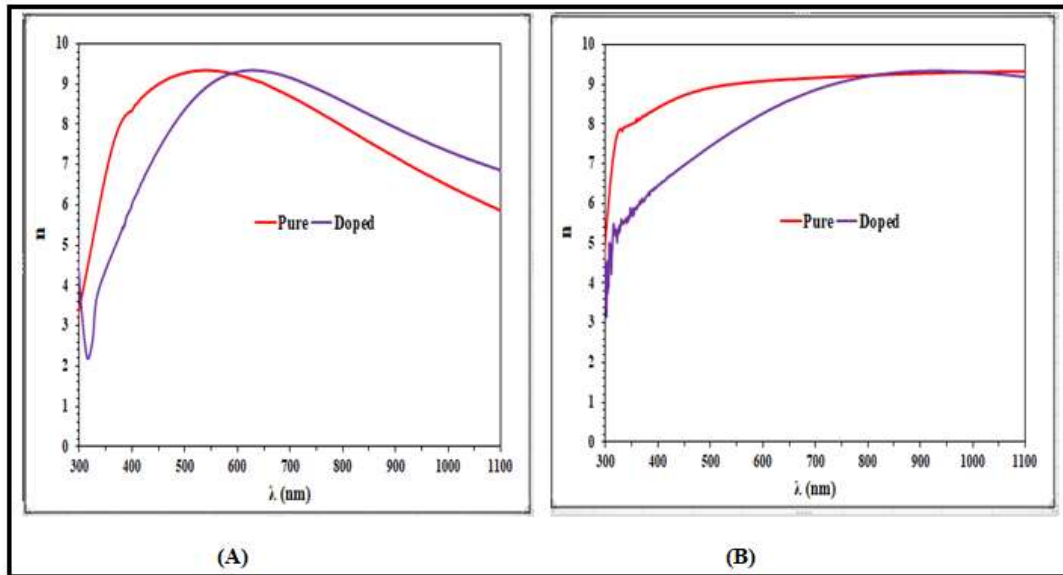


Figure 11. Shows The refractive index of (A): pure SnO₂ nanocrystalline & SnO₂:Cu annealed at 300°C, and (B): pure SnO₂ nanocrystalline & SnO₂:Cu annealed at 500°C.

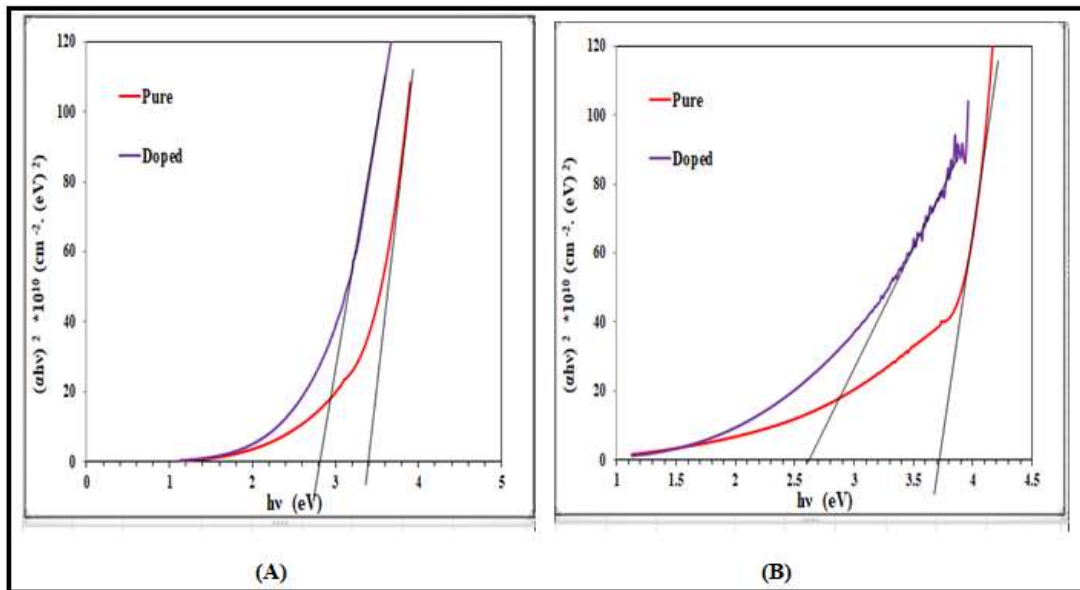


Figure 12. Shows optical energy gap of (A): pure SnO₂ nanocrystalline & SnO₂:Cu annealed at 300°C, and (B): pure SnO₂ nanocrystalline & SnO₂:Cu annealed at 500°C.

4. Conclusions

In this paper, thin films of tin oxide doped with 2% copper were fabricated by spray pyrolysis method, the XRD pattern showed that the SnO₂ nanocrystals are tetragonal crystals and that the particle size was within the nanosize of the pure and doped substance. AFM and FESEM result It proved that the particles were spherical and of different sizes. As for the results of UV spectroscopy, it was found that the energy gap decreases with the increase of doping. As for the effect of the annealing temperature on the permeability and absorbance, it was

shown that the permeability decreases with the increase in the annealing temperature, while the absorbance increases. From the compositional results, energy gap values, absorbance increase, roughness, and other results obtained, it is clear that the prepared samples can be used in different applications such as solar cells and gas sensors.

References

- [1] M. J. Glover, E. Jones, K. L. Masconi, M. J. Sweeting, S. G. Thompson, and S. Collaborators, "Discrete event simulation for decision modeling in health care: lessons from abdominal aortic aneurysm screening," *Med. Decis. Mak.*, vol. 38, no. 4, pp. 439–451, 2018.
- [2] N. K. Abass, A. A. Baker, A. J. Cadhum, and N. A. Mahmood, "Structural and optical properties of SnS₂: Cu thin films prepared by chemical spray pyrolysis," *Baghdad Sci. J.*, vol. 7, no. 1, pp. 57–68, 2010.
- [3] R. L. Mishra, S. K. Mishra, and S. G. Prakash, "optical and gas sensing characteristics of tin oxide nano-crystalline thin film," *J. Ovonic Res.*, vol. 5, no. 4, 2009.
- [4] N. Sankara Subramanian, B. Santhi, S. Sundareswaran, and K. S. Venkatakrishnan, "Studies on spray deposited SnO₂, Pd: SnO₂ and F: SnO₂ thin films for gas sensor applications," *Synth. React. Inorganic, Met. Nano-Metal Chem.*, vol. 36, no. 1, pp. 131–135, 2006.
- [5] P. M. Mwathe et al., "Influence of surface passivation on optical properties of spray pyrolysis deposited Pd-F: SnO₂," 2014.
- [6] B. V. Odari et al., "Optoelectronic properties of F-co-doped PTO thin films deposited by spray pyrolysis," *Am. J. Mater. Sci.*, vol. 3, no. 4, pp. 91–99, 2013.
- [7] A. E. Rakhshani, Y. Makdisi, and H. A. Ramazaniyan, "Electronic and optical properties of fluorine-doped tin oxide films," *J. Appl. Phys.*, vol. 83, no. 2, pp. 1049–1057, 1998.
- [8] P. M. Mwathe et al., "Effect of Annealing and Surface Passivation on Doped SnO₂ Thin Films Prepared by Spray Pyrolysis Technique," *Adv. Mater.*, vol. 4, no. 3, p. 51, 2015.
- [9] P. M. Mwathe et al., "Surface passivation effect on CO₂ sensitivity of spray pyrolysis deposited Pd-F: SnO₂ thin film gas sensor," *Adv. Mater.*, vol. 3, no. 5, pp. 38–44, 2014.
- [10] F. de Moure-Flores et al., "SnO₂: F thin films deposited by RF magnetron sputtering: effect of the SnF₂ amount in the target on the physical properties," *Rev. Mex. física*, vol. 59, no. 4, pp. 335–338, 2013.
- [11] S. M. Al-Delaimy and R. Basheer, "Annealing Effect on Structural and Optical Properties of SnO₂ Thin Films," *J. Educ. Sci.*, vol. 21, no. 2, pp. 92–102, 2008.
- [12] N. F. Habubi, Z. M. Abood, and A. N. Algamel, "Molar concentration Effects on the Optical and Structural Properties of nanostructural SnO₂ Thin Films," *Int. Lett. Chem. Phys. Astron.*, vol. 65, p. 81, 2016.
- [13] V. K. Jain, P. Kumar, and Y. K. Vijay, "Preparation of nanostructure ZnO-SnO₂ thin films for optoelectronic properties and post annealing influence," *Int. J. Mater. Metall. Eng.*, vol. 6, no. 12, pp. 1192–1194, 2012.
- [14] A. A. Yadav, E. U. Masumdar, A. V Moholkar, K. Y. Rajpure, and C. H. Bhosale, "Gas sensing of fluorine doped tin oxide thin films prepared by spray pyrolysis," *J. Sens. Transducers*, vol. 92, pp. 55–60, 2008.
- [15] N. Jebbari, N. Kamoun, and R. Bennaceur, "Effect of SnCl₄ concentration on F: SnO₂, deposited by chemical spray pyrolysis," in the proceedings of International Renewable Energy Congress, 2010, pp. 276–279.
- [16] P. S. Patil, "Versatility of chemical spray pyrolysis technique," *Mater. Chem. Phys.*, vol. 59, no. 3, pp. 185–198, 1999.

- [17] E. Elangovan and K. Ramamurthi, "Optoelectronic properties of spray deposited SnO₂: F thin films for window materials in solar cells," *J. Optoelectron. Adv. Mater.*, vol. 5, no. 1, pp. 45–54, 2003.
- [18] P. Karthick, D. Vijayanarayanan, S. Suja, M. Sridharan, and K. Jeyadheepan, "Opto-electronic properties of fluorine doped tin oxide films deposited by nebulized spray pyrolysis method," *Asian J Appl Sci*, vol. 8, no. 4, pp. 259–268, 2015.
- [19] Nachaithong T, Sikam P, Moontragoon P, Thongbai P, Kaewmaraya T, Ikonc Z. The study of optical and colossal dielectric properties of (Cu, Ga)-doped ZnO nanoparticles," *Eng Appl Sci Res.*, 2021;48(6):759–65.
- [20] F. I. M. Fazli et al., "the effects of annealing temperature on properties of Aluminium-doped tin oxide (AL/SnO₂), thin films deposited by spray pyrolysis deposited (SPD) method," 2006.
- [21] Mohamed MB, Abdel-Kader MH, "Effect of annealed ZnS nanoparticles on the structural and optical properties of PVA polymer nanocomposite," *Mater Chem Phys.*, vol. 241, pp. 122285, 2020.
- [22] B. Yulianto, G. GUMILARA, D. W. ZULHENDRIA, and N. L. W. Septiani, "Preparation of SnO₂ Thin Film Nanostructure for CO Gas Sensor Using Ultrasonic Spray Pyrolysis and Chemical Bath Deposition Technique.," *Acta Phys. Pol. A.*, vol. 131, no. 3, 2017.
- [23] C. W. Su, L. Lu, and M. O. Lai, "Recrystallization and grain growth of deformed magnesium alloy," *Philos. Mag.*, vol. 88, no. 2, pp. 181–200, 2008.
- [24] Pham V-T, Fang T-H, "Influences of grain size, alloy composition, and temperature on mechanical characteristics of Si_{100-x}Gex alloys during indentation process," *Mater Sci Semicond Process.*, vol. 123, pp. 105568, 2021.
- [25] S. A. Makee, A. A. M. Shehab, and A. M. Rehima, "Effect of Annealing Temperatures on the Structural and Optical Properties of ZnO and ZnO: Al Thin Films Prepared By Thermal Evaporation Technique," *Ibn AL-Haitham J. Pure Appl. Sci.*, vol. 26, no. 3, pp. 153–160, 2017.
- [26] P. Sagar, P. K. Shishodia, R. M. Mehra, H. Okada, A. Wakahara, and A. Yoshida, "Photoluminescence and absorption in sol-gel-derived ZnO films," *J. Lumin.*, vol. 126, no. 2, pp. 800–806, 2007.
- [27] S. Ray, P. S. Gupta, and G. Singh, "electrical and optical properties of sol-gel prepared pd-doped SnO₂ thin films: effect of multiple layers and its use as room temperature methane gas sensor.," *J. Ovonic Res.*, vol. 6, no. 2, 2010.